



## BALLISTIC PERFORMANCE OF A QUENCHED AND TEMPERED STEEL AGAINST 7.62MM CALIBRE PROJECTILE

O. M. Sanusi<sup>1</sup> and J. O. Akindapo<sup>2</sup>

<sup>1</sup> RESEARCH AND DEVELOPMENT CENTRE, DEFENCE INDUSTRIES CORPORATION OF NIGERIA, NIGERIA

<sup>2</sup> DEPARTMENT OF MECHANICAL ENGINEERING, NIGERIAN DEFENSE ACADEMY, KADUNA, NIGERIA

*E-mail addresses:* <sup>1</sup> [sanuthwale@gmail.com](mailto:sanuthwale@gmail.com), <sup>2</sup> [jacobakindapo@gmail.com](mailto:jacobakindapo@gmail.com)

### ABSTRACT

*In this research effort, ballistic performance of a quenched and tempered steel was investigated. Low alloy steel was selected where austenization, quenching and finally tempering at 600°C were applied to it. Thereafter, the heat-treated steel was shot with armour piercing 7.62 mm calibre and the occurrence of failure, after the interaction between the projectile and the steel, was investigated. The shot was performed at zero degree (0°) obliquity with a projectile velocity of 830m/s. After the shot, microstructural and fractographical examinations were carried out on the sample taken from the perforated region using scanning electron microscopes to determine the matrix phase and secondary phases. It was observed that the steels had tempered martensitic-bainitic matrices after heat treatments; a crater was formed on the front side of the steel; deformed and transformed adiabatic shear bands had an effect on the crack formation and propagation in the matrix; and perforation mode of the steels was typical petalling.*

**Keywords:** Projectile, petalling, ballistic impact, perforation, quenched & tempered steel

### 1. INTRODUCTION

Metal and its alloys, ceramics, polymers and composite materials can be used as armour materials in structural protection technology. The conceptions such as hardness, strength and toughness are the main features for the ballistic performance of a given material. Recently, defence industry has been trying to find out materials having excellent ballistic performance under any defined threat and all attempts focus on the design of new alloys/materials, single or multi composite systems, processing techniques and secondary treatments. For this purpose, the search and development studies on blast and penetration-resistant materials (BPRMs) are very popular in many disciplines [1-3].

Many metals make great BPRMs and the most common ones include steels (ferrous alloys), aluminium and titanium alloys. The two purposes of metals in structural protection are stated as follows: protection against fragments and maintaining structural integrity. Metals are highly useful in protecting structures against explosions because of

their inherent strength, toughness and energy absorption capability [1, 3]. The most known alloy as protection material is the armour steel. The only armour grade steel, which is currently used for structural applications, is the rolled homogenous armour (RHA) [1, 4]. Armour steel should have properties which include: high resistance to perforation and ballistic impacts; easy fabricability; adequate fatigue and wear resistance under service conditions.

Hardness is an important feature for the materials used for armour strategy [1-3]. Sangoy and others.[5] reported that high hardness of given armour steel directly determines the ballistic performance and perforation mode. Many studies on the ballistic impact behaviour of the steels revealed that relationship exists between hardness of the steel matrix and ballistic performance of the steel after dynamic impact. In addition, toughness is another critical property for a given armour material under a dynamic attack of projectiles having high kinetic energies. It is generally considered that armour steels

having high toughness will be very useful to resist ballistic impacts without being fractured. As it is well known, alloying and also heat treatments affect the toughness of the materials [2, 5 - 7].

This research was initiated in order to harness the potentials of Nigeria’s commercially available steel in developing armour plates via application of heat treatments consisting of austenisation, quenching and tempering. These heat treatment processes are observed on the frequently imported armour steel. Thereafter, the research studied the performance of the treated plate against armour piercing 7.62 mm calibre and the occurrence of failure in order to evaluate its feasibility in ballistic applications.

**2. EXPERIMENTAL PROCEDURE**

**2.1. Materials and Methods**

Table 1 shows the chemical composition of the commercial 10mm thick steel plate used in this study. 300 x 1000mm plate was prepared from the sheet for the ballistic test while the steel has an original alloy design to develop a new armour steel according to MIL-DTL-46100D/E [8-9].

Table 2 shows the heat treatment conditions applied on the experimental steel at the Heat and Surface Treatment Factory, Defence Industries Corporation of Nigerian (DICON), Kaduna. As it may be seen, the heat treatment (thermo-mechanical process) consists of austenization, quenching, and finally tempering in accordance with the MIL-DTL-46100D/E standards and with conventional armour steels [8-11].

All mechanical properties are represented in Table 3. It should be noted that Notched impact test was carried out at a temperature of -20°C.

*Table 1: The chemical composition (Wt-%) of the experimental steel*

Metal	% composition
C	0.24
Mn	0.19
Si	0.18
Ni	0.04
Co	2.35
Cr	1.4
Mo	0.5
Nb	0.08
V	0.08
Ti	0.01
P	0.01
B	0.01
S	0.01
Fe	Balance

*Table 2: The heat treatment conditions of the experimental steel*

Austenisation	1000 °C, 45 minutes
Quenching Medium	Water
Tempering	600 °C, 45 minutes

*Table 3: The mechanical properties of the experimental steel*

Tensile Strength (MPa)	1270
Harness (HRC)	341
Enlongation (%)	0.1
Impact toughness(J)	8.20

**2.2. Ballistic Test**

The ballistic test methodology is explained in detail in the MILSTD-662F [12]. A shot was performed by a 7.62 mm (calibre) armour piercing projectile with a velocity of 830 m/s from 30 m range. Table 4 shows the shot condition used for the experimental steel.

The 300 x 1000mm steel plate was prepared for the ballistic test which was carried out at Quality Control (shooting range), Defence Industries Corporation of Nigeria, Nigeria.

The ballistic performance, according to MIL-A-12560 and MIL-A-46100 standards, was reported by Atapeket *al.* in many studies, as a function of the applied heat treatments [1,13,14].

*Table 4: The shot condition for the experimental steel*

Distance	30 m
Shot angle	0°
Projectile type	7.62-51 armour piercing
Ambient temp./ Rel. humidity	21 °C/ 42 per cent

**2.3. Metallographic and Fractographic Examinations**

Samples taken from the experimental steels before and after the shots (from the perforated zones) were prepared by metallographical methods. All samples were prepared by grinding with 320, 600 and 1000 mesh size Si Ca brasives respectively. Then, the ground surface was polished with 3µm diamond solution. Etching was carried out with nital (3% of HNO<sub>3</sub>) to characterize the microstructure. PhenomProX Scanning Electron Microscope (at Ahmadu Bello University, Zaria, Nigeria) was used for both metallographic and fractographic examinations.

**3. RESULTS AND DISCUSSION**

**3.1. Microstructure of the quenched/tempered steel plate sample**

Generally, the microstructure of steel determines its physical and chemical properties under loading

condition. For armour steel, the matrix having martensitic/bainitic tempered martensitic-bainitic structure determines the ballistic performance. This is achieved after application of austenitisation and then quenching on low carbon and alloyed steel [5, 15]. Several studies emphasized that a martensitic/bainitic structure and morphology of these phases, content of retained austenite, austenization/tempering conditions directly affected the final failure mode or ballistic performance of heat treated steels used as armour [16-19]. On the other hand, selection of appropriate material and processing conditions and resulting microstructural and mechanical properties immensely affect the protective characteristics of the material under any dynamic loading.

Figure 1 shows the scanning electron microscope image of the etched experimental steel. The matrix typically exhibits a tempered bainite microstructure. The regions in gray contrast denote the ferritic matrix which refers to decomposed and coarsened lath due to tempering. Intensive precipitates having higher hardness compared to the matrix are seen in dark contrast, since they have lower reflective index. The grain boundaries are clearly seen in the microstructures. As it is well-known, all transformations start at these regions and the austenite grain size must be controlled for higher mechanical requirements [20].

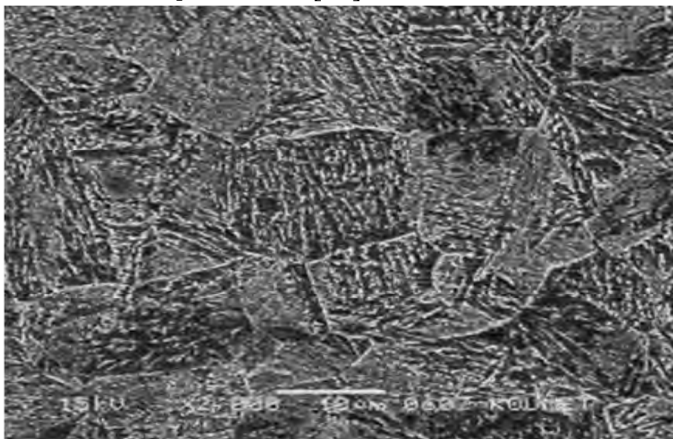


Figure 1: Scanning electron microscope image of the tempered bainite microstructure

### 3.2. Macro and micro examinations on the projectile-perforated steel sample

After the ballistic shot, the cross-section of perforated zone is shown in Figure 2. At the entrance of the steel sheet, a crater (with a certain depth and length) forms at the beginning of the penetration. It is inevitable that the armour steel fails in cleavage fracture, which means no plastic deformation, in the case of the first interaction with a projectile having high strike velocity

and kinetic energy. At high strain rates (e.g., some metal cutting operations, projectile impacts with high rates of 100-3600 m/s or fracture due to blast), materials exhibit local deformation known as adiabatic shear [21, 22]. Adiabatic shear bands (ASBs) are formed as a result of a thermomechanical instability due to the presence of a local inhomogeneity, including local deformation and heating. If the thermal conductivity of the material is not sufficient to conduct the generated heat away, deformation becomes unstable and is localised on surfaces of very small thickness ( $\sim 10 \mu\text{m}$  to  $50 \mu\text{m}$ ).

This situation is compatible in the interaction of a projectile on a steel target. High temperatures can form because of high friction of the projectile during penetration. On microscopic examination, adiabatic shear bands appear as narrow bands in which cracks can propagate, indicating catastrophic failure of the material [23].

A fractographic analysis was done on perforated steel to determine the crater regions and output line. All examinations are given in Figures 3 and 4. The perforated steel was examined in two folds: In the first stage of the examination, it is observed that crater regions have bright and smooth faces indicating no plastic deformation (Figure 3). In the interaction of a projectile with the material, kinetic energy is consumed as fracture, deformation and heat energy in general. For a given armour material, blunting of a sharp projectile causes a decrease in the efficiency of ammunition. Therefore, a requirement for the ballistic performance can be provided. Consuming of energy as fracture plays an important role at this stage. A typical adiabatic shear band formed in the matrix of the experimental steel due to high strain by impact loading of the projectile. Formed adiabatic shear bands play a role on the nucleation and subsequently on the propagation of the crack [1,2].

In the second stage of the examination, it is observed that the steel has a plastic deformation capability under ballistic impact due to yielding in the output line of projectile. In this stage, the perforation mode of the steel is typical petalling (Figure 4).

The formation of deformed and transformed ASBs in the experimental steel after shot has a similarity to the study on AISI 4340 steel, commonly used in dynamic loading test such as high strain rate test, by Bassim, et al [23]. On the other hand, adiabatic shear band results in several perforation modes in armour steel (e.g., plugging and discing type fracture).

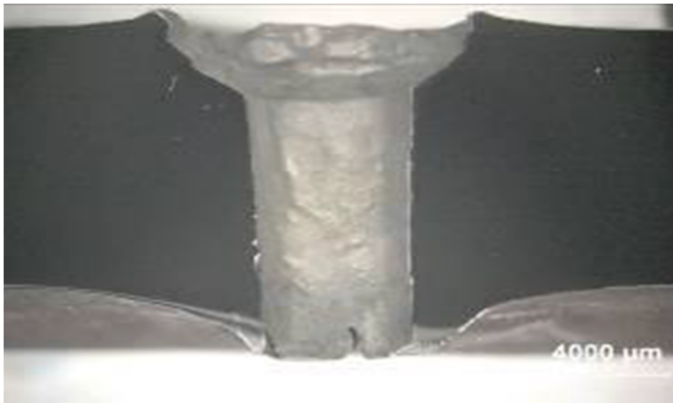


Figure 2: Cross-section of the projectile-perforated steel sample (projectile entrance was from top and left through the lower end of the picture)

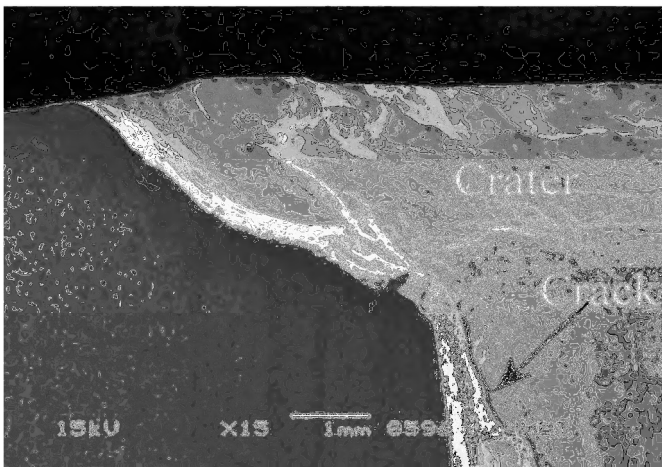


Figure 3: Formation of crater (perforation at entrance) due to penetration of the projectile and the formation of crack because of ASBs followed by the regions of cleavage fracture and abrasive wear

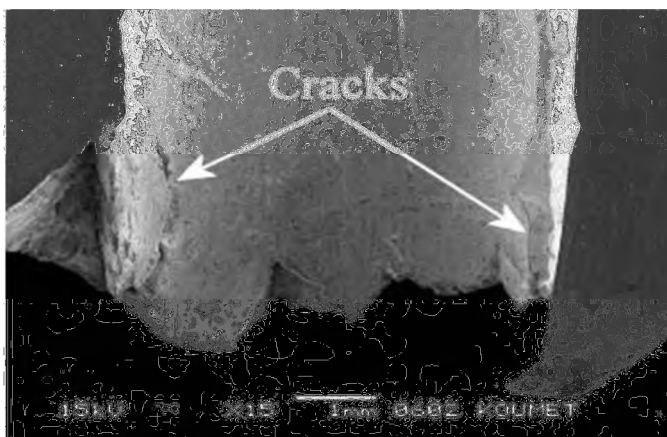


Figure 4: The formation of cracks and petalling at the line of departure (failure formed in the steel after perforation)

#### 4. CONCLUSIONS

In this study, the ballistic behaviour of quenched and tempered steel has been investigated. After applied heat treatments consisting of austenisation, quenching and finally tempering, a matrix with 34HRC hardness was obtained. The changes in the microstructure, the

deformation and perforation mode of the steel were considered after the shot performed at zero degree with a projectile velocity of 830 m/s. The experimental steel has a tempered bainitic matrix due to applied heat treatments stated in Table 2.

The matrix exhibits the decomposition of the bainite, which is transformed from austenite grain boundaries by quenching, after tempering. The original matrix was deformed due to over impact loading by the penetration of the projectile. The adiabatic shear bands and cracks near the direction of penetration were observed. The adiabatic shear bands were formed as a result of the local heterogeneities, deformation and thermomechanical instability due to overheating by the friction of projectile moving along the steel. The narrow bands were characterised as white bands after metallographical examinations and the cracks were formed on or close to the bands. This results in a failure as perforation or fragmentation in the steel.

The fractographical examinations on the cross-section of perforated zone after shot support the microstructural characterisation and the formation in the micro-level (e.g., the deformation, the formation of adiabatic shear bands, and cracks). The cleavage type fracture due to strain hardened microstructure close to penetration direction was observed. The formation of the abrasive wear is inevitable under loading and motion of the projectile. The cracks due to adiabatic shear bands were determined both at the first step of the projectile interaction with the steel and also at the line of departure. There is certain elongation from target material to outside at the line of departure and this elongation displays the intensive plastic deformation. The perforation is a typical petalling, that is breakings from target material/steel.

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